

Solid-State Electrochemical Micromachining

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Surface microstructuring of substrates plays a significant role in microelectronics and optoelectronics. The process has recently been extended to the fabrication of chemical systems such as microreactors¹ and micropumps.² The textured surfaces are fabricated using one of two general approaches. One is the topographic deposition of desired materials onto the surface.³ This is the “build-up” type approach, in which the surface is constructed from the molecular or atomic scale. Micromachining belongs to the other category and can be considered to be a “shaving off” type texturing method. Electrochemical micromachining (EM), which works by local dissolution of a conducting substrate (metals, semiconductors) under an applied anodic bias in solution, is one of the most widely used methods because it requires simple equipment and enables more rapid etching than other techniques such as ion beam milling and laser abrasion. However, a liquid electrolyte, which is difficult to handle, is required as a conducting medium between the two electrodes. In addition, high-resolution patterning requires masking. The latter problem is solved by the combined use of position selective laser and ion beam irradiation to draw a minute pattern,^{4–6} or the application of an ultrashort voltage pulse.^{7–9} In this communication, we propose a simple, novel route for solid-state micromachining using an anodic electrochemical reaction at the microcontact between the ion conducting microelectrode and metal substrate. More concretely, the metal substrate is locally incorporated into the ion conductor in the form of metal ions via the microcontact

under a dc bias. The substrate is micromachined as a result of the continuous application of an electric field, which eliminates the need for wet-processing. The present method, which we shall refer to as solid-state electrochemical micromachining (SSEM), can easily control the machining size and depth by adjusting the contact area or other EM parameters. Moreover, scanning the microelectrode under an applied electric field results in a fine-patterned surface. The following sections will focus on the fundamental mechanism behind SSEM and present some experimental results.

The model for ion migration in the SSEM system is depicted in Figure 1a. The β'' -Al₂O₃ (typically, Na- β'' -Al₂O₃), which conducts a variety of metal ions in the interlayer of spinel blocks, was employed as an ion conductor. The solid-state electrochemical cell consists of an Ag plate (cathode)/pyramid-like Na- β'' -Al₂O₃ microelectrode¹⁰/target metal substrate (thickness: 0.5 mm; M: Ag or Zn; anode) system. To reduce the mechanical stress at the microcontact to a minimum and remain constant, the experimental setup was devised with only the weight of the microelectrode (ca. 0.02 g) to the metal substrate. When a dc electric field is applied to the cell, the local region of the M anode near the solid–solid interface is electrochemically oxidized to Mⁿ⁺,¹¹ which, in turn, is injected into the Na- β'' -Al₂O₃ via the microcontact. Na⁺ migrates through the β'' -Al₂O₃ and is deposited as Na metal at the Ag cathode/Na- β'' -Al₂O₃ interface. Na metal immediately reacts with the O₂ and CO₂ in the air to form Na₂CO₃. Thus, the electrolysis mechanism corresponds to the electrochemical substitution of Mⁿ⁺ for Na⁺ in the β'' -Al₂O₃. As a result, the M anode electrochemically dissolves as Mⁿ⁺ into β'' -Al₂O₃. Since the contact radius at the Na- β'' -Al₂O₃/M interface is on the order of 10 μ m,¹² a position selective dissolution occurs at the microcontact, and thus SSEM is achieved.

SSEM was carried out under galvanostatic conditions. Figure 2 shows the time dependence of applied voltage during a typical constant-current electrolysis run without (a) and with (b) tracing the β'' -Al₂O₃ microelectrode along the surface of the Ag plate. When the microelectrode was fixed on the Ag surface (Figure 2a), the applied voltage gradually decreased except during the initial stage of electrolysis (<20 min). Others have reported that the voltage response in the

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- (10) The commercial Na- β'' -Al₂O₃ (diameter: 7 mm; thickness: 2 mm) or a Ag- β'' -Al₂O₃ polycrystal was used as an ion conductor. Ag- β'' -Al₂O₃ was prepared by ion exchange of Na- β'' -Al₂O₃ in a AgNO₃ molten salt over 24 h. Complete exchange of Ag for Na was confirmed by electron probe microanalysis (EPMA) in the cross section of β'' -Al₂O₃ ceramics after exchange. The obtained β'' -Al₂O₃ was polished with emery paper to form a pyramid-like microelectrode as shown in Figure 1b.
- (11) Thévenin-Annequin, C.; Levy, M.; Pagnier, T. *Solid State Ionics* **1995**, *80*, 175.
- (12) The contact area between the microelectrode and metal substrate in Figure 1a was estimated by scanning electron microscopy (SEM) observation of the surface depressions produced by pressing the microelectrodes onto the metal surface under actual experimental conditions. Assuming a hemispherical microcontact between the microelectrode and the metal, the contact radius was estimated to be on the order of 10 μ m. This value was slightly altered by the apex configuration of the hand-polished microelectrode and the processing temperature.

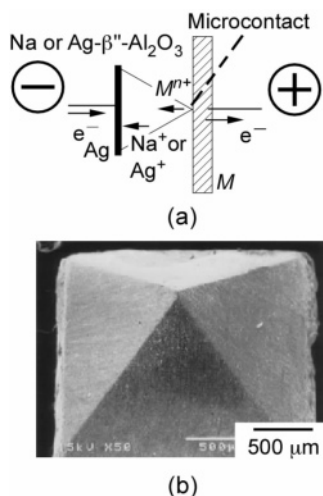


Figure 1. (a) Model for ion migration during SSEM of metal substrate. (b) SEM image of typical β'' - Al_2O_3 microelectrode.

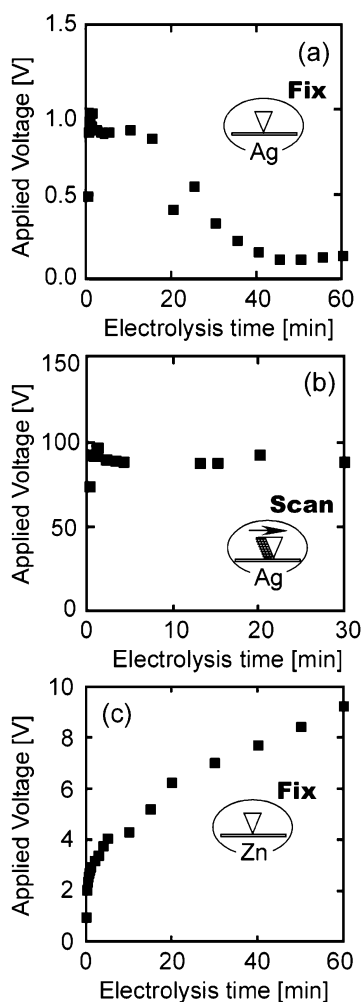


Figure 2. Typical time evolution of applied voltage under constant current electrolysis; (a) Ag micromachining using fixed microelectrode ($100\ \mu\text{A}$, $873\ \text{K}$). (b) Ag micromachining on scanning microelectrode (scan rate: $1\ \mu\text{m/s}$, $50\ \mu\text{A}$, $673\ \text{K}$). (c) Zn micromachining using fixed microelectrode at $0.1\ \mu\text{A}$ and $623\ \text{K}$.

ion conducting microelectrode technique depends on the electrical resistance only in the small region near the microcontact^{13,14} because the current density at the Ag

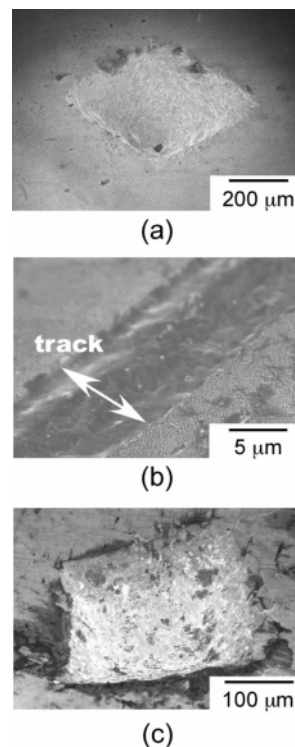


Figure 3. (a) Surface depression of Ag plate after applying $100\ \mu\text{A}$ for 20 min at $873\ \text{K}$ using fixed microelectrode. (b) Line-patterned Ag surface under $50\ \mu\text{A}$ at $673\ \text{K}$ on scanning microelectrode ($1\ \mu\text{m/s}$). (c) Zn surface after point-machining at $0.1\ \mu\text{A}$ and $623\ \text{K}$ for 10 h.

(cathode)/ β'' - Al_2O_3 would be negligible compared with that at the microcontact. Taking into account the fact that the type of mobile cations (Na^+ or Ag^+) in β'' - Al_2O_3 is independent of the applied voltage, the voltage changes in Figures 2a and 2b seemed to be mainly influenced by the electrochemical reaction at the microcontact (Ag^+/Ag). Thus, the decrease in voltage suggests that the contact area between the pyramid-like microelectrode and Ag plate increased with electrolysis time. That is, the microelectrode was embedded in the Ag anode. By contrast, when the microelectrode was scanned during electrolysis, the applied voltage remained almost constant throughout the duration of electrolysis (Figure 2b), except for some variations that may have been due to the surface roughness or the slight change in temperature. In this case, the contact area of the microcontact was maintained constant, and machining always proceeded on the original Ag surface.

SSEM of the Ag plate was performed using a fixed $\text{Na-}\beta''$ - Al_2O_3 microelectrode. Ag was detected in the cross section of the $\text{Na-}\beta''$ - Al_2O_3 by EPMA measurement after electrolysis. Thus, the ion migration mechanism proposed above actually proceeded during the electrolysis. Figure 3a shows an SEM image¹⁵ of the Ag surface after micromachining under a constant current of $100\ \mu\text{A}$ for 20 min at $873\ \text{K}$. The black dot around the center of the image was the maximum depth position (about $120\ \mu\text{m}$), which the top of the microelectrode reached just after micromachining.¹⁶ The photograph indicates that the Ag plate was drilled

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(15) To better characterize the surface morphology, all SEM samples were inclining by 35° with respect to the perpendicular electron beam axis.

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according to the apex form of the microelectrode, i.e., in the inverse pyramid form. Naturally, the depth was changed by the applied electricity or the microelectrode shape. The quantitative assessment¹⁷ revealed that the current efficiency of the SSEM for the Ag plate (100 μ A, 873 K, 1 h) using a fixed Na- β'' -Al₂O₃ microelectrode was about 40%. The current loss may be caused by the sparks generated at the inhomogeneous contact between the microelectrode and the metal substrate. Consequently, SSEM was accomplished by using the ion conducting microelectrode. In conventional EM with no masking, the machining size is always larger than that of the counter electrode¹⁸ because of the presence of the electrolyte solution as a conducting medium. On the other hand, since SSEM utilizes direct ion migration via a solid–solid microcontact, the machining dimension is almost identical to the effective contact area. Moreover, Figure 3a suggests that the apex configuration (including the surface roughness) of the microelectrode can be directly transferred to the surface of the work piece.

To groove the Ag plate on a micrometer scale, the Ag- β'' -Al₂O₃ microelectrode equipped with an XYZ three-dimensional auto stage was scanned along the Ag surface under an electric field. Figure 3b is the SEM image of the Ag surface after scanning the microelectrode in a single direction at 50 μ A and 623 K (scan rate: 1 μ m/s). The track was produced along the pathway of the microelectrode, and its width and depth were about 10 and 1 μ m, respectively. The step observed in the groove may be due to the surface roughness of the sides of the microelectrode. As in the case of a fixed microelectrode, the machining size (width and depth) could be controlled by current and/or scan rate. Since the microelectrode was attached to the automated XYZ stage, we could draw various groove patterns on the metal Ag surface, not only simple structures (points and lines) but also more complex forms.

Divalent cations can be substituted into the interlayers of the β'' -Al₂O₃ crystal structure alongside the monovalent ions, in which case it behaves as a divalent cation conductor.¹⁹ Thus, a variety of different kinds of metal substrates can presumably be used as a work piece. SSEM of a Zn plate, which supplies the divalent Zn²⁺ and is relatively inert toward surface oxidation at high temperature, was investigated using a fixed Na- β'' -Al₂O₃ microelectrode. Figure 2c shows the typical time evolution of applied voltage under a current of 0.1 μ A at 623 K below the melting point of zinc (692 K). In contrast to the case of the Ag plate, applied voltage tended to increase monotonically with time, even if the microelec-

trode was fixed. This may be due to the much lower ionic conductivity of divalent cations (Zn²⁺) as compared to that of monovalent cations (Na⁺) in the β'' -Al₂O₃ structure.²⁰ Figure 3c shows the photograph of a Zn surface after 10 h of SSEM under the same conditions as in Figure 2c. A pyramid-like depression was observed on the Zn surface. This indicates that Zn²⁺ could be removed from Zn metal by using SSEM. A variety of (mono-tetravalent) metal ion conductors have been developed in recent years by many different research groups.^{21,22} This means that SSEM can be performed on a whole host of metal substrates, which is an essential merit of the present technique.

Ion conducting microelectrodes have been widely used in the field of solid-state ionics,^{23,24} which deals with fast ion conduction in solids. In particular, microelectrodes play an important role in the measurement of local conductivity in ceramic materials²⁵ and transference numbers in mixed electronic-ionic conductors.^{13,14} By contrast, we have used ion conducting microelectrodes as metal source of pinpoint ion doping into solid materials.^{26,27} Moreover, it was proved that the microelectrode is an effective tool for the electrochemical patterning of the metal distribution in the surface of or inside glass on a micrometer scale.^{28,29} The SSEM method may revolutionize such conventional applications based on novel aspects of the ion conducting microelectrode. SSEM has many advantages over conventional EM. For instance, since it is a solid-state process, no solutions are required, and the metal surface can be directly patterned in almost any configuration desired. SSEM has a number of other merits commonly noted with respect to EM: easy control of machining size and three-dimensional structure by electrochemical parameters and microelectrode configuration. On the other hand, some demerits such as the slow etching rate, the roughness of the machined surface, and the low current efficiency are also found out in the present technique. Therefore, in the future, these problems will soon be solved by advances in electrode fabrication or optimization of electrolysis conditions.

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- (16) The crystal structure and surface morphology of the β'' -Al₂O₃ microelectrode were not changed after the SSEM according to the XRD analysis and the SEM observation. However, the β'' -Al₂O₃ microelectrode often adhered to the metal substrate after applying high voltage, and its apex broke off the microelectrode in some cases.
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